Multiscale Materials Modeling At the University of Tennessee

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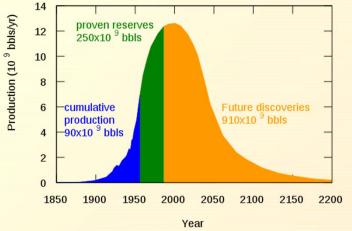


MSE 101 University of Tennessee, Knoxville April 9, 2014

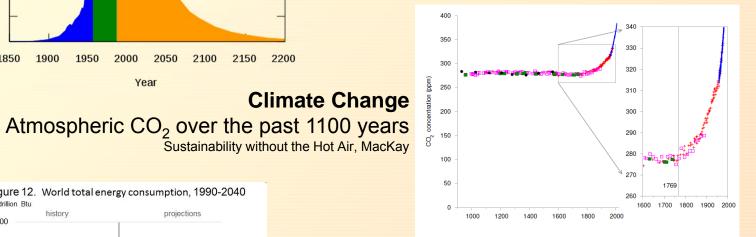
Multiscale Materials Modeler

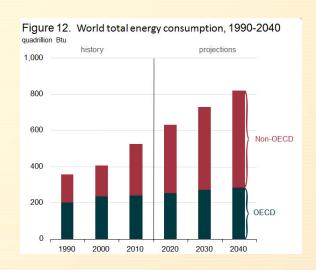


Renewable Energy: The Defining Challenge of Your Generation



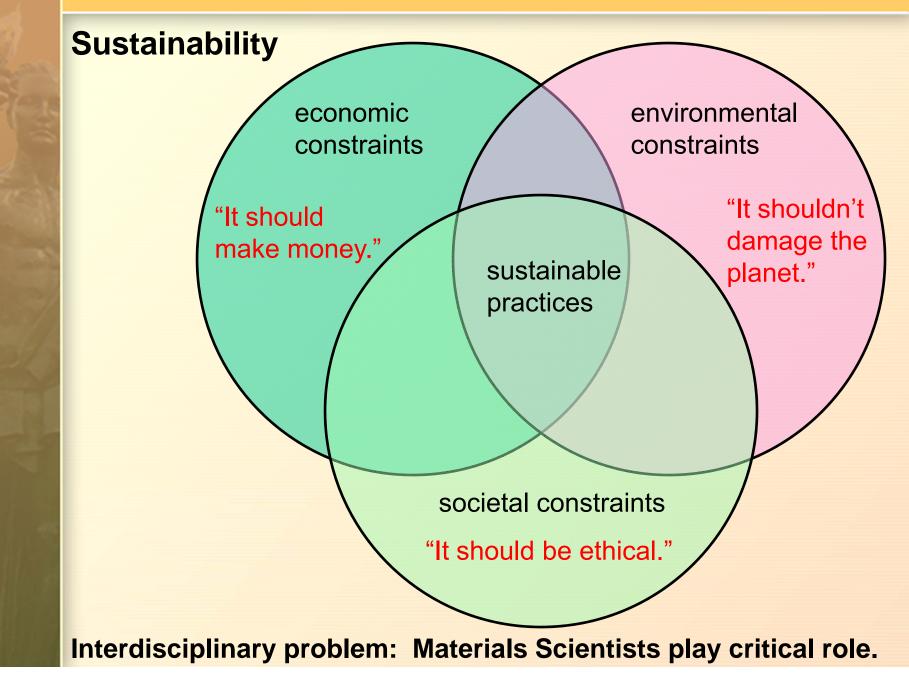
Peak Oil Fossil fuels are a finite resource http://en.wikipedia.org/wiki/Peak oil



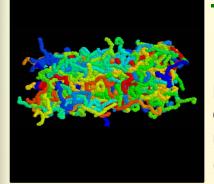


Global Energy Demand is Rising

http://www.eia.gov/forecasts/ieo/world.cfm



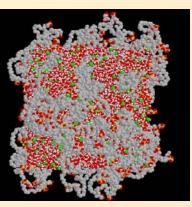
Apply simulation tools to develop structure/property relationships

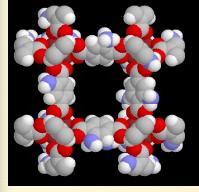


polymeric materials

polymers at equilibrium and under flow (PE, PET)

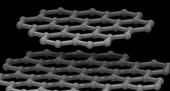
polymer electrolyte membranes (PEMs) in fuel cells

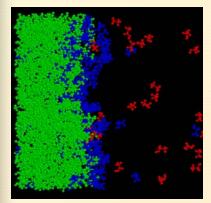




nanoporous materials

hydrogen sorption in metal organic frameworks (MOFs) bio-derived, nanostructured battery anodes

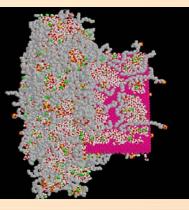


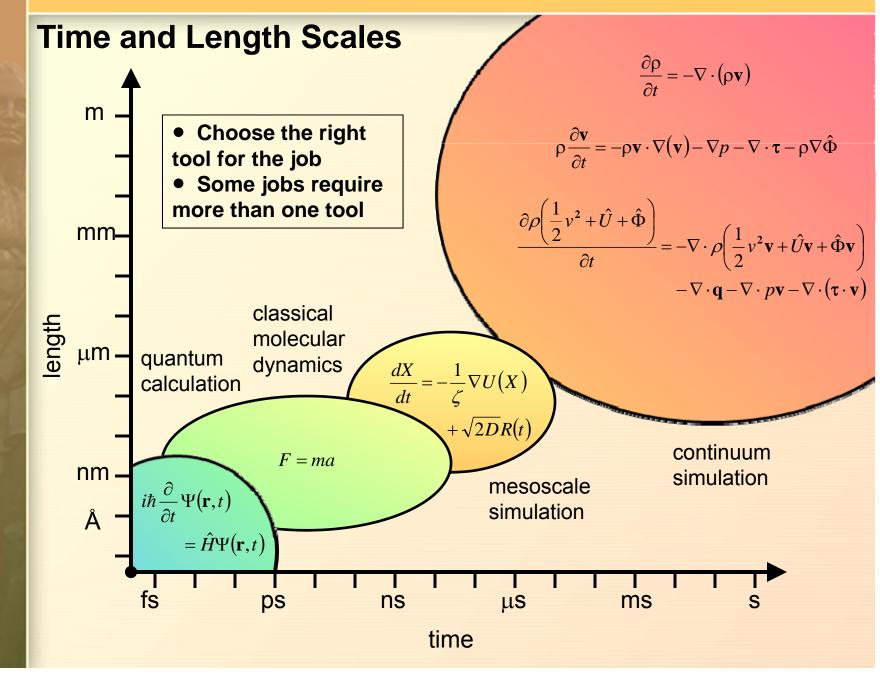


interfacial systems

near critical vapor-liquid interface structure

fuel cell electrode/ electrolyte interfaces

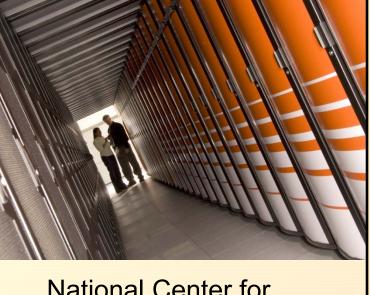




Collaboration with Oak Ridge National Laboratory

OAK RIDGE NATIONAL LABORATORY

Managed by UT-Battelle for the Department of Energy



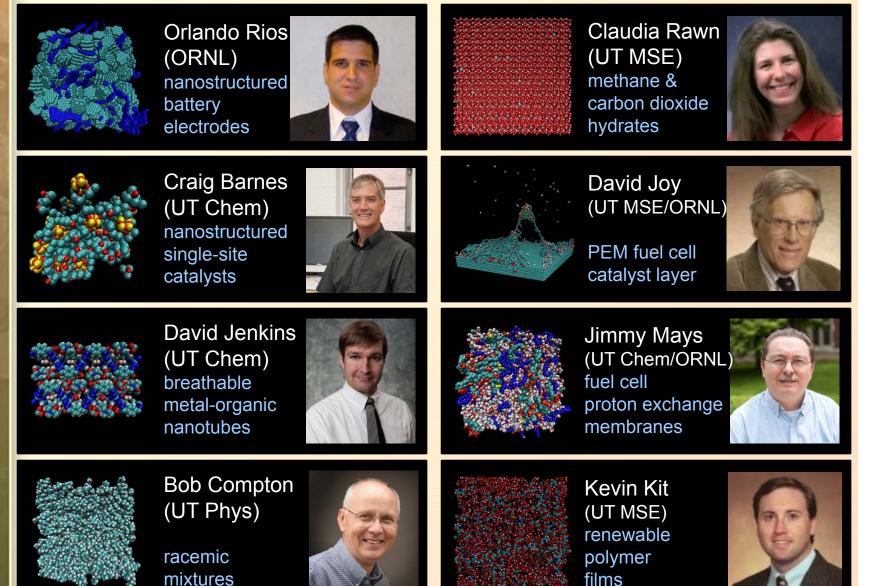
National Center for Computational Science

Today the computing resources of the NCCS are among the fastest in the world, able to perform more than 119 trillion calculations per second.

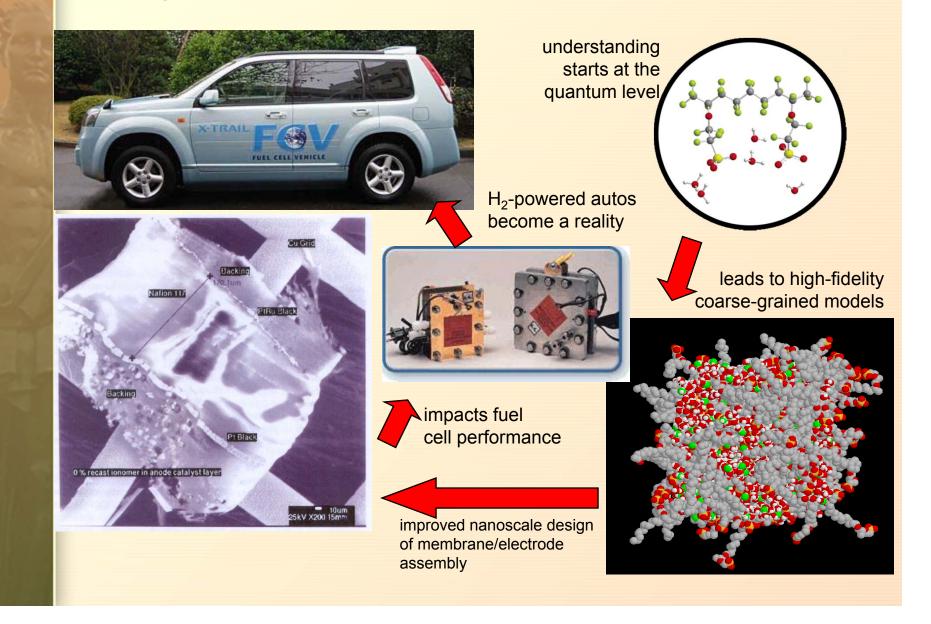
To solve systems of ODEs (largest system thus far is several million), we use the massively parallel supercomputers at ORNL.

These resources are available to researchers at UT through discretionary accounts of the program directors.

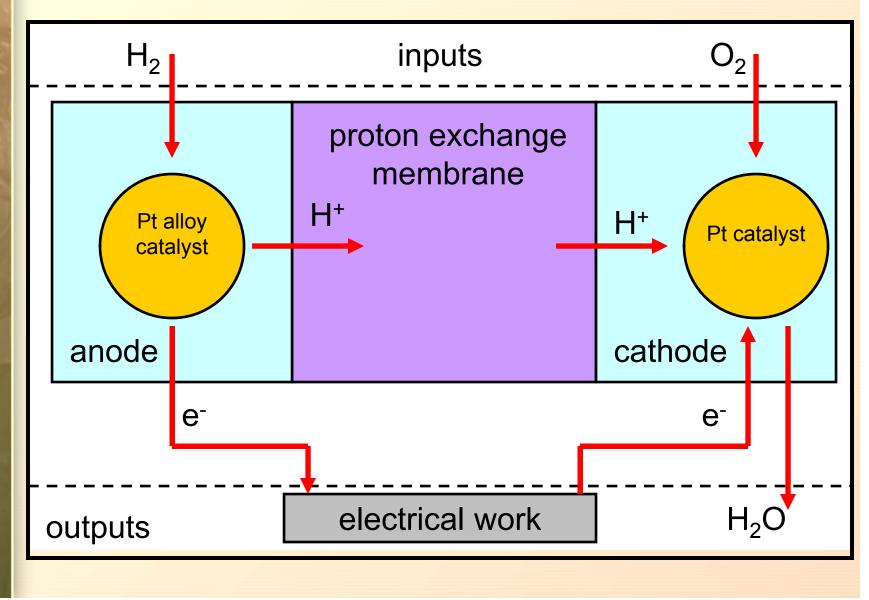
A Complementary Tool: Experimental Collaborators (2013)



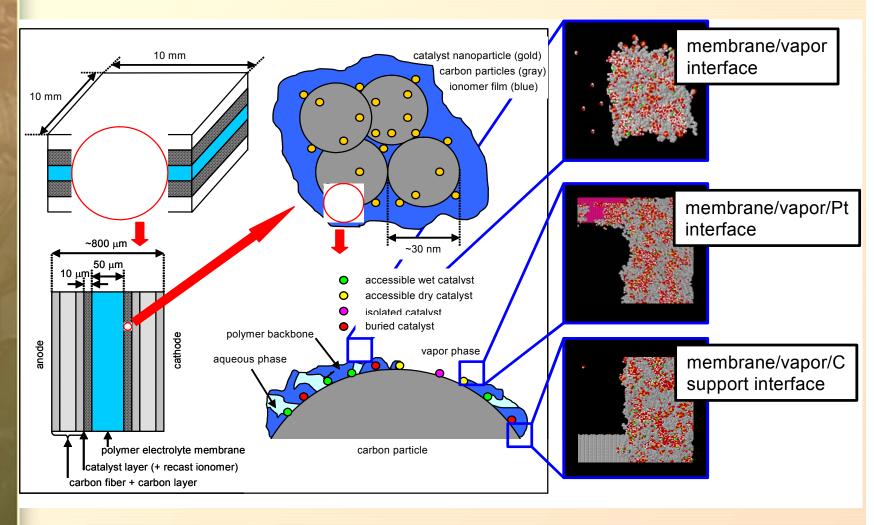
Moving toward fuel cell-powered vehicles



how fuel cells work: conceptual level



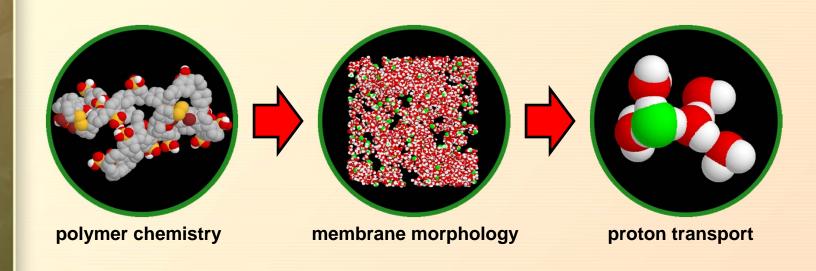
Fuel Cells are composed of a number of nanostructured materials: carbon fibers, catalyst nanoparticles, polymeric electrolyte membranes.



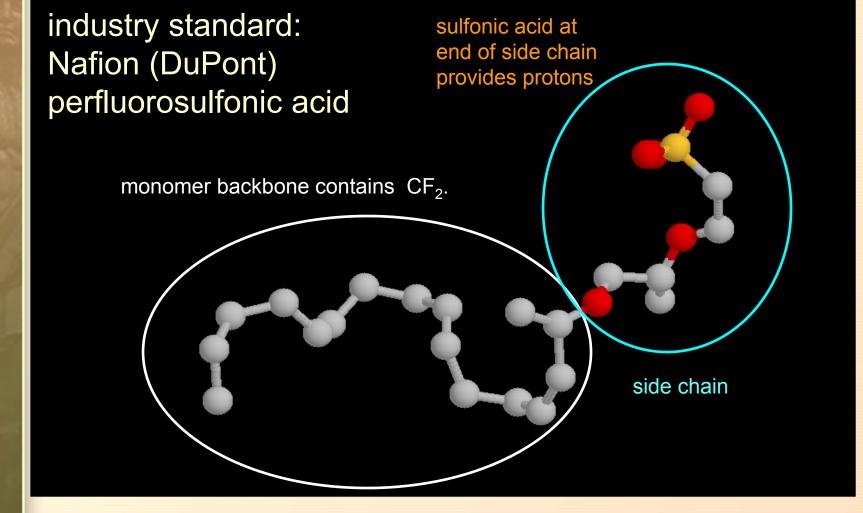
A membrane electrode assembly from the macroscale to the molecular scale.

Research Questions

- 1. What is the relationship between polymer chemistry and the morphology of the hydrated membrane?
- 2. What is the relationship between the morphology of the hydrated membrane and the membrane transport properties?



proton exchange membranes are polymer electrolytes

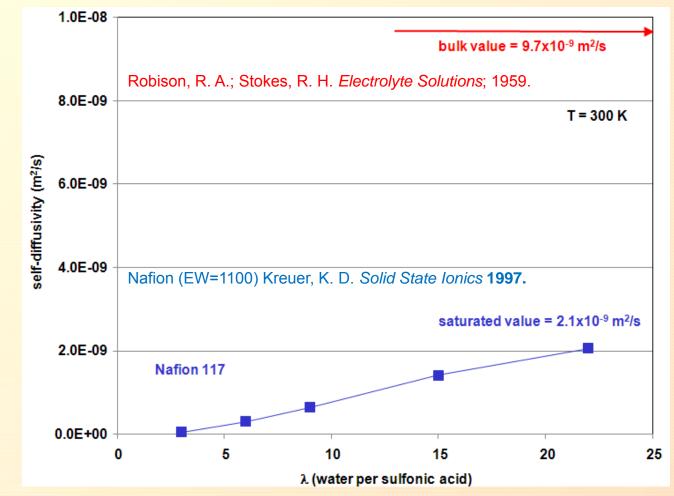


 CF_2 = gray, O = red, S = orange, cation not shown.

Motivation for new proton exchange membranes

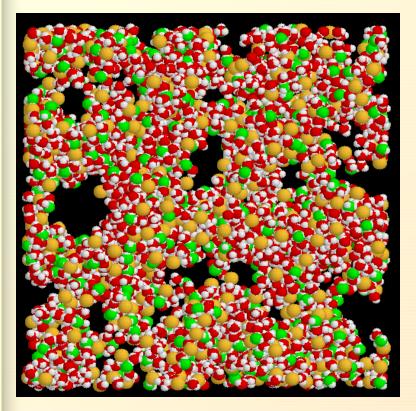
- Lower Cost reduce noble metal (Pt or Pt alloy) catalyst content
- Higher Operating Temperature
 - catalyst
 - higher activity
 - less susceptible to poisoning due to fuel impurities (CO)
 - membrane
 - dries out
 - conductivity drops
- High Temperature (120 °C) proton exchange membranes
 - retain moisture at higher temperatures
 - maintain high conductivity at lower water content

Proton Transport in Bulk Water and PEM Experimental Measurements



Even at saturation, the self-diffusivity of charge in Nafion is 22% of that in bulk water.

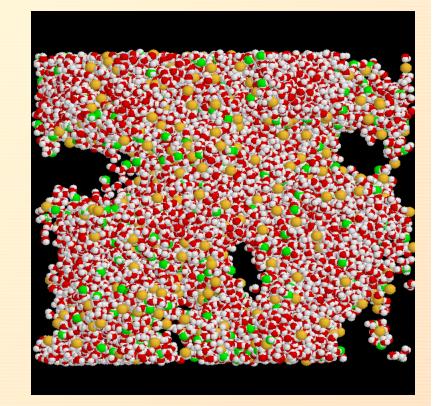
PEM morphology is a function of water content



Nafion (EW = 1144) λ = 6 H₂O/HSO₃ small aqueous channels

Nafion (EW = 1144) λ = 22 H₂O/HSO₃ much larger aqueous channels

As the membrane becomes better hydrated, the channels in the aqueous domain become larger and better connected, resulting in higher conductivity. (The challenge to finding high-temperature membranes is to find one that can retain moisture at elevated temperatures.)



Determination of Diffusivities from MD Simulation

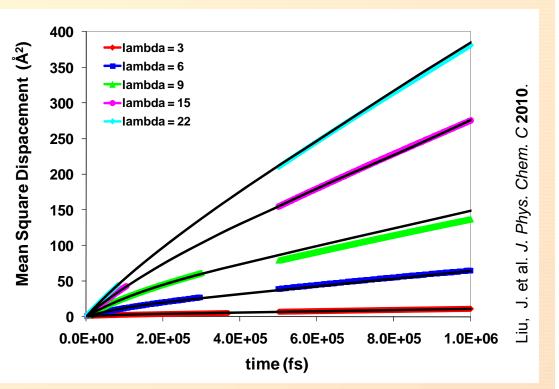
Einstein Relation – long time slope of mean square displacement to observation time

$$D = \lim_{\tau \to \infty} \frac{MSD}{2d\tau} = \lim_{\tau \to \infty} \frac{\left\langle \left[r_i(t+\tau) - r_i(t) \right]^2 \right\rangle}{2d\tau} \quad \begin{array}{c} \text{position of} \\ \text{particle } i \text{ a} \\ \text{time } t \end{array}$$

Einstein Relation works well for bulk systems.

But for simulation in PEMs, we can't reach the long-time limit required by Einstein relation.

MD simulations alone are not long enough.

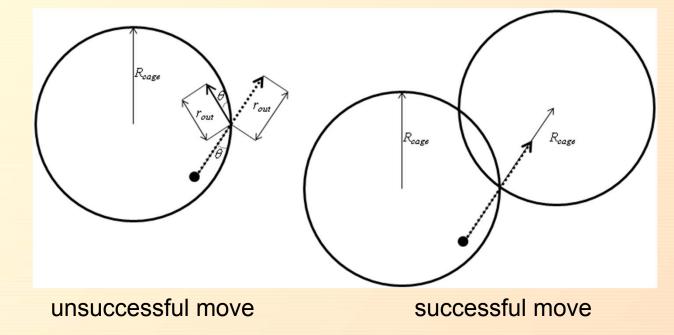


MSDs don't reach the long-time (linear) regime.

Confined Random Walk Simulation

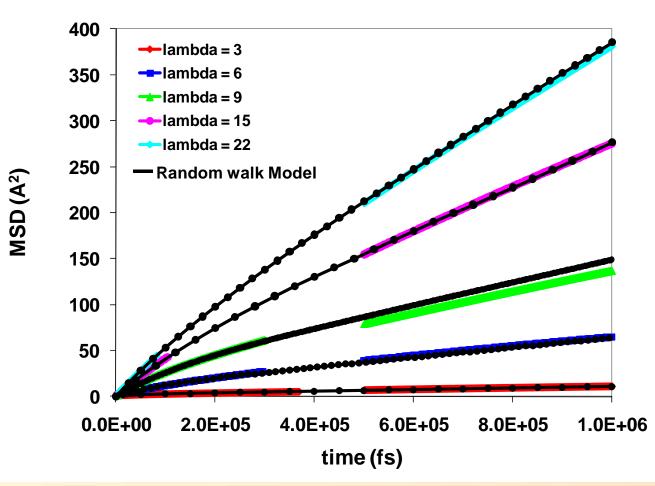
Mesoscale Model

- non-interacting point particles (no energies, no forces)
- sample velocities from a Maxwell-Boltzmann distribution
- two parameters
 - cage size
 - cage-to-cage hopping probability
- parameters fit to MSD from Molecular Dynamics Simulation
- runs on a laptop in a few minutes



Phys. Rev. E, 83(1) 2011 article Ojha, M., Keffer, Esai Selvan, M., Xiong, R., Egami, T D.J., Nicholson, D.M., # 011120. E.N Calvo-Muñoz,

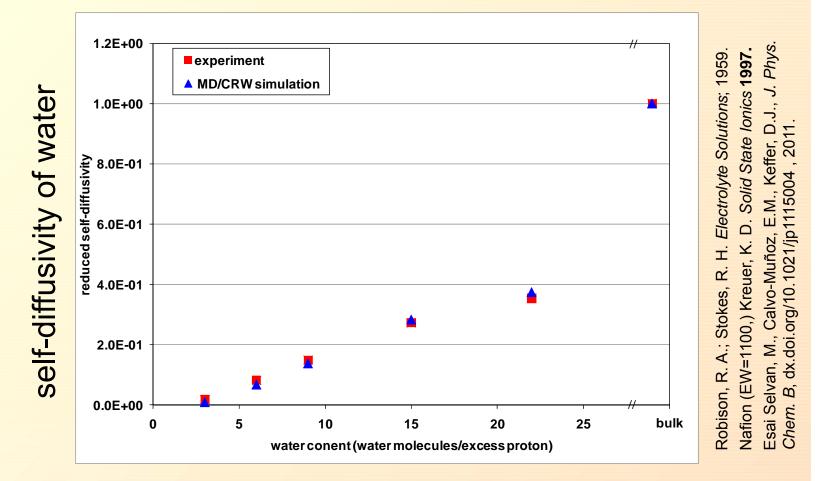
Couple MD with Confined Random Walk (CRW) Theory



- Fit MD results (1 ns) to Confined Random Walk (CRW) Theory.
- Extend Mean Square Displacement to long-time limit (100 ns).
- Extract water diffusivity.

Phys. Rev. E, 83(1) 2011 article Ojha, M., Keffer, Esai Selvan, M., Xiong, R., D.J., Nicholson, D.M., Egami, T # 011120. E.M., Calvo-Muñoz,

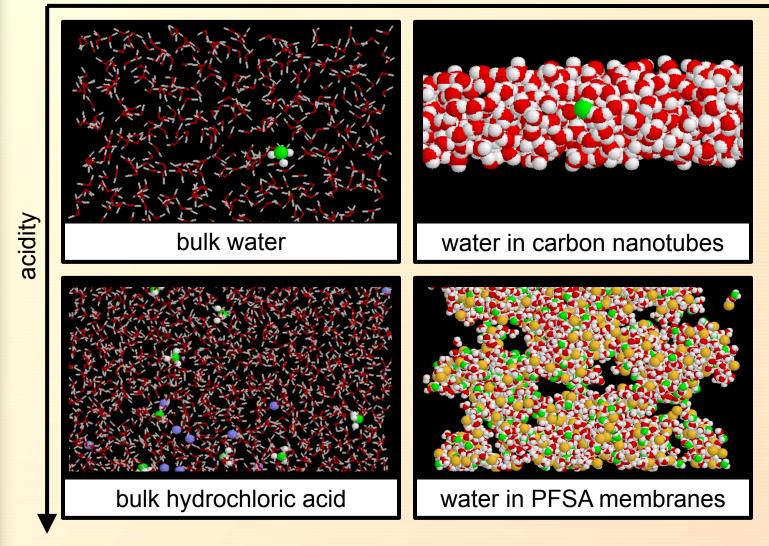
Comparison of MD/CRW Simulation with Experiment



- Excellent agreement between simulation and experiment for water diffusivity as a function of water content
- Can we predict the self-diffusivity of water without computationally expensive simulations?

Acidity and Confinement Effects on Proton Mobility

confinement

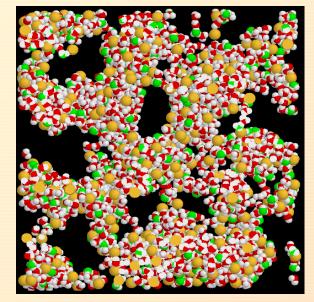


Water Mobility in Bulk Systems – Effect of Connectivity

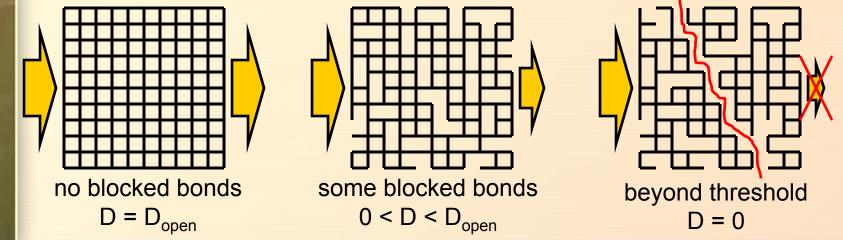
Invoke Percolation Theory to account for connectivity of aqueous domain within PEM and obtain effective diffusivity.

$$\int_{0}^{\infty} \frac{D_{eff} - D}{\left(\frac{z}{2} - 1\right)D_{eff} + D} g(D)dD = 0$$

$$g(D) = p_{EMA} \delta(D - D_b) + (1 - p_{EMA}) \delta(D - D_o)$$

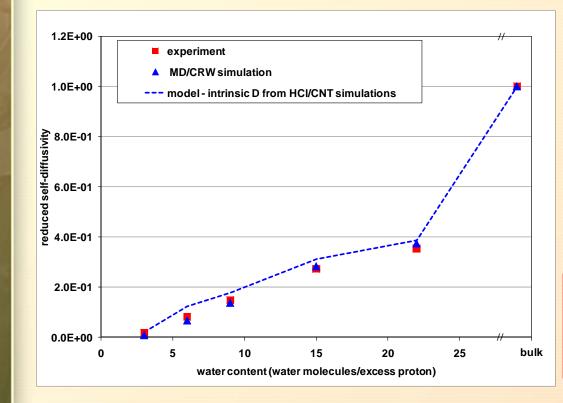


Percolation theory relates the effective diffusivity to the fraction of bonds that are blocked to diffusion.



Structure-Based Analytical Prediction of Self-diffusivity

- Acidity characterized by concentration of H₃O⁺ in aqueous domain (exponential fit of HCI data)
- Confinement characterized by interfacial surface area (exponential fit of carbon nanotube data)
- Connectivity characterized by percolation theory (fit theory to MD/CRW water diffusivity in PEMs)



Excellent agreement of theory with both simulation and experiment.

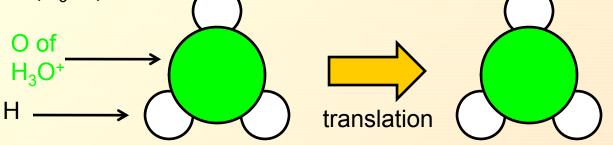
Theory uses only structural information to predict transport property.

Water is solved! What about charge transport?

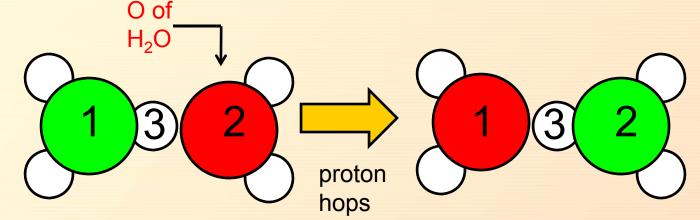
Esai Selvan, M., Calvo-Muñoz, E.M., Keffer, D.J., *J. Phys Chem. B* **115**(12) 2011 pp 3052–3061.

Proton Transport – Two Mechanisms

Vehicular diffusion: change in position of center of mass of hydronium ion (H_3O^+)



Structural diffusion (proton shuttling): passing of protons from water molecule to the next (a chemical reaction involving the breaking of a covalent bond)



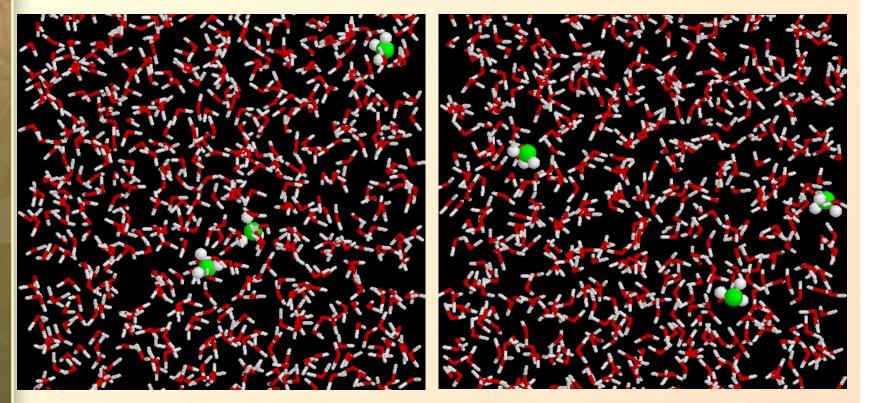
In bulk water, structural diffusivity is about 70% of total diffusivity.

RMD In Water

Proton Diffusion in Bulk Water

Non - Reactive System

Reactive System

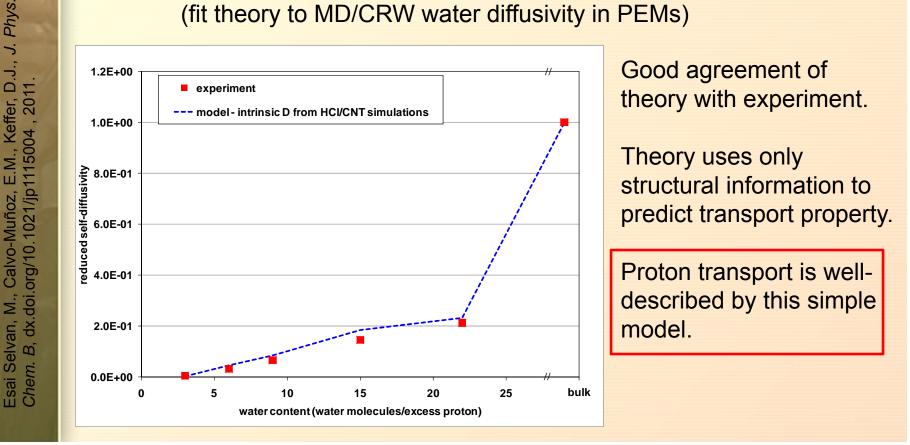


Vehicular Diffusion

Structural and Vehicular Diffusion

Structure-Based Analytical Prediction of Self-diffusivity

- Acidity characterized by concentration of H₃O⁺ in aqueous domain (exponential fit of HCl data)
- Confinement characterized by interfacial surface area (exponential fit of carbon nanotube data)
- Connectivity characterized by percolation theory (fit theory to MD/CRW water diffusivity in PEMs)

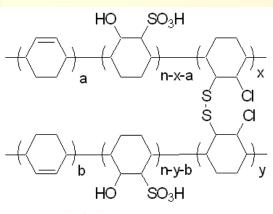


Generalizability of the Analytical Approach

Determine the extent to which a small set of structural descriptors can be used to describe transport in nanostructured materials

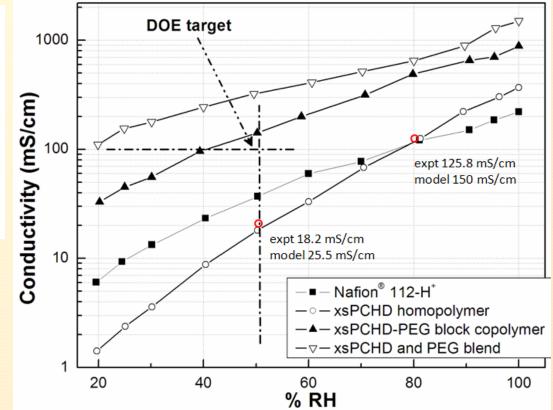
- other proton exchange membranes composed of homopolymers
- other proton exchange membranes with additives or blends
- Li+ ion transport in nanostructured electrodes

cross-linked and sulfonated Poly(1,3-cyclohexadiene)

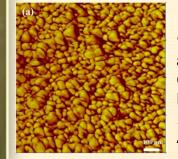


xsPCHD homopolymer

Percolation theory approach works for xsPCHD membrane as well.



Wang, Q., Suraweera, N.S., Keffer, D.J., Deng, S., Mays, J.W., *Macromolecules*, DOI: 10.1021/ma300383z 2012.



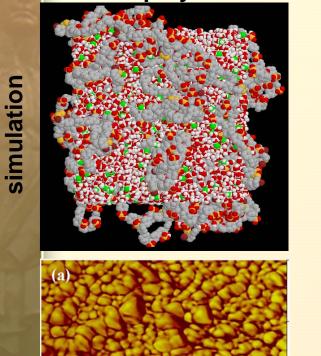
"Polymer Electrolyte Membranes with Enhanced Proton Conductivities at Low Relative Humidity based on Polymer Blends and Block Copolymers of Poly(1,3-cyclohexadiene) and Polyethylene Glycol By Suxiang Deng, Amol Nalawade, Mohammad K. Hassan, Kenneth A. Mauritz, and Jimmy W. Mays* Advanced Materials, 2012, under review.

cross-linked and sulfonated Poly(1,3-cyclohexadiene)

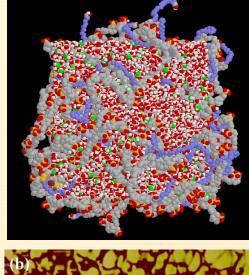
xs-PCHD homopolymer

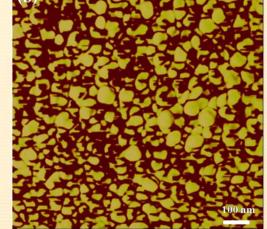
xs-PCHD/PEG copolymer

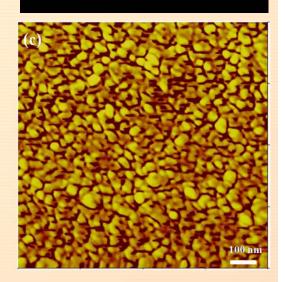
xs-PCHD/PEG blend



AFM

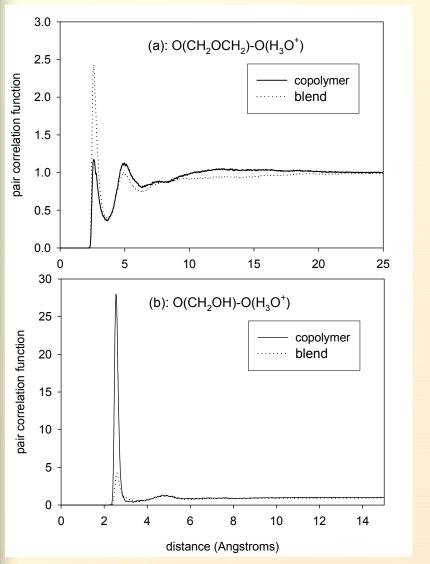


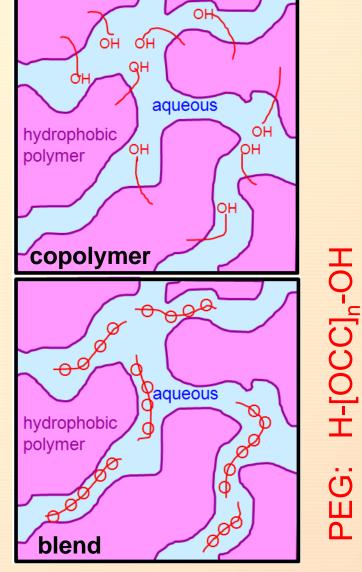




xs-PCHD/PEG blend shows best performance.

xsPCHD/PEG copolymer and blend: structure





Distribution of PEG in membrane is very different for copolymer and blend.

Acknowledgments



Office of Basic Energy Sciences

This work is supported by the United States Department of Energy Office of Basic Energy Science through grant number DE-FG02-05ER15723.

OAK RIDGE NATIONAL LABORATORY Managed by UT Battelle for the Department of Energy

Access to the massively parallel machines at Oak Ridge National Laboratory through the UT Computational Science Initiative.

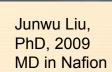
All xsPCHD experimental data from Suxiang Deng & Prof. Jimmy Mays, UTK Chemistry.



Qifei Wang, PhD 2011, xsPCHD



Myvizhi Esai Selvan PhD, 2010 Reactive MD





Nethika Suraweera PhD, 2012 Vol & Area Analysis

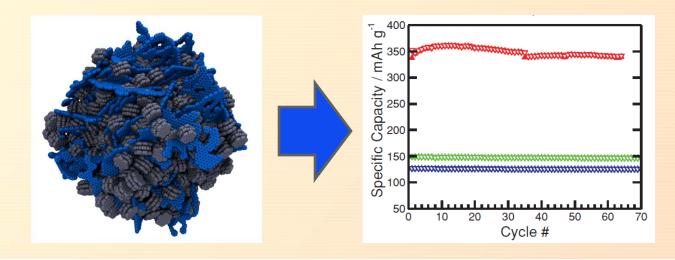
Elisa Calvo-Munoz undergraduate Random Walks

Multiscale Modeling of Carbon Composite Electrodes From Renewable Materials

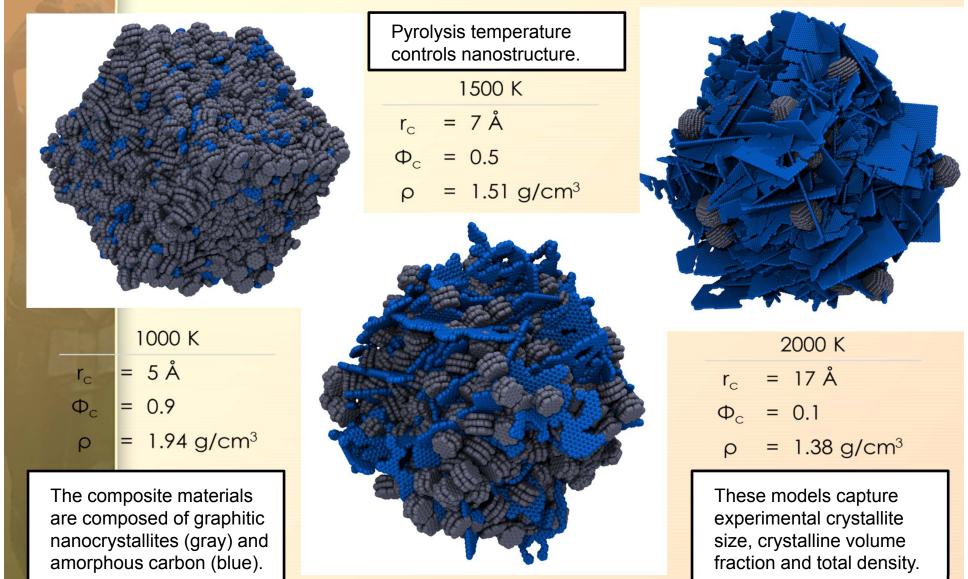
David J. Keffer¹, Nicholas W. McNutt, Khorgolkhuu Odbadrakh & Orlando Rios² ¹University of Tennessee & ²Oak Ridge National Laboratory

dkeffer@utk.edu

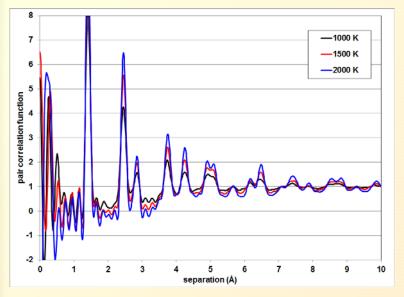
Objective: The objective of this work is to understand the molecular-level mechanisms responsible for the exceptionally high ion storage and fast charging and discharging rates observed in the novel lignin-based carbon composite electrodes synthesized by Rios at ORNL. This knowledge can be used to further guide development of improved materials for battery electrode applications.



Molecular Models of Experimentally Synthesized Composites

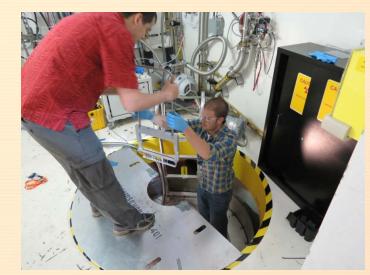


Neutron Diffraction from NOMAD

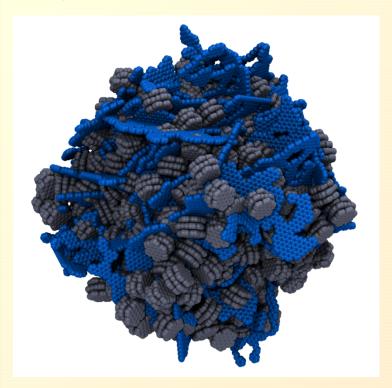


NOMAD is a high-flux, medium-resolution diffractometer that uses a large bandwidth of neutron energies and extensive detector coverage to carry out structural determinations of local order in crystalline and amorphous materials.





Interpretation of Nomad Data



		500	К
r _c	=	7 Å	
Φ _c	=	0.5	
ρ	=	1.51	g/cm ³

Process

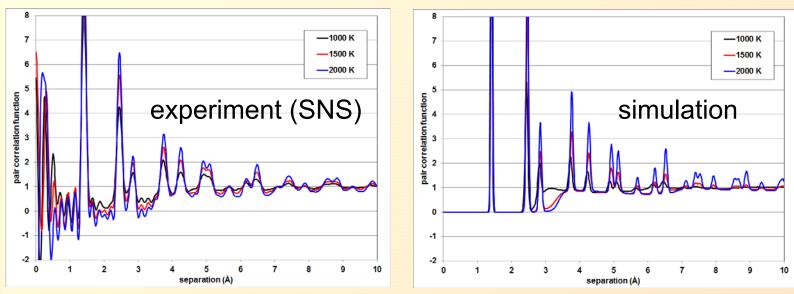
Modelers use their knowledge and imagination to hypothesize structures.

Perform MD simulations.

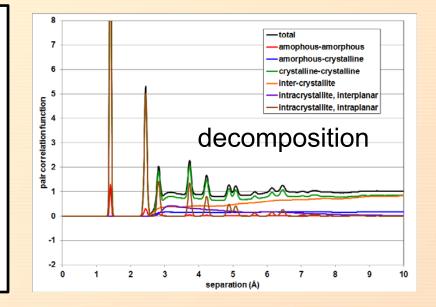
Generate pair correlation functions (PCFs).

Compare simulated and experimental PCFs.

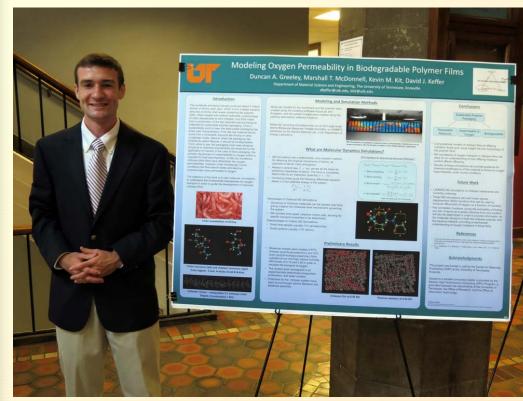
Composite Models Provide Interpretation of Neutron Data



Neutron diffraction data from the SNS (top left) are difficult to interpret for even partially amorphous materials. However, clear trends are seen with respect to pyrolysis temperature. Pair correlation functions (PCFs) for the corresponding models (top right) provide clean, unambiguous data. Moreover, the simulated PCFs can be completely decomposed (right) to reveal the structural origins of all features in the spectra, providing clear understanding of the experimental data from the SNS.



Undergraduates Perform Research in MSE at UT





Duncan Greeley performs MD simulations of oxygen transport in chitosan films to provide insight into biodegradable plastics made from renewable resources. (2013)

Conclusions

• The search for renewable energy sources and systems is the defining challenge of your generation.

• Materials Scientists & Engineers play a critical role in this search for sustainability.

 Students in the Materials Science & Engineering Department at the University of Tennessee are performing state-of-the-art research using the world's best supercomputers and neutron sources to develop new materials for alternative energy systems.

- Multiscale Materials Modeling is a complementary tool to experiment, providing unique insight.
- Experimental/Computational collaborations are fruitful and fun!

Questions?

